# STRUCTURAL STABILITY OF TERNARY ANTIMONIDES $T_{10}Sb_{5}T'$ (T=Ti, Zr, Hf, T'=V, Cr, Mn, Fe, Co, Ni, Cu, Pd, Pt, Rh)

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#### Abstract

The crystal and thermodynamic properties of  $T_{10}Sb_5T'$  ternary compounds where T=Ti, Zr, Hf and T' = V, Cr, Mn, Fe, Co, Ni, Cu, Pd, Pt, Rh have been investigated by means of first principle calculations. The structure of these compounds is considered as a ternary variant of the  $W_5Si_3$  type. Three possible ordered structures have been investigated in the present work. The two structures in which the Sb and T' atoms alternate along chains parallel to the c axis are the most stable ones. The calculated structural parameters are in good agreement with the experimental data. Both electronic and size effects allow to explain the stability of the ternary  $T_{10}Sb_5T'$  compounds.

**Keywords:** DFT calculations; Enthalpies of formation; W<sub>s</sub>Si, ternary variant; Nb<sub>s</sub>Sn<sub>s</sub>Si-type.

## ESTABILIDADE ESTRUTURAL DE COMPOSTOS TERNÁRIOS DE ANTIMONIO $T_{10}Sb_{5}T'$ (T=Ti, Zr, Hf, T'=V, Cr, Mn, Fe, Co, Ni, Cu, Pd, Pt, Rh)

#### Resumo

As propriedades termodinâmicas e estruturas cristalinas dos compostos ternários do tipo  $T_{10}Sb_5T'$  onde T=Ti, Zr, Hf and T' = V, Cr, Mn, Fe, Co, Ni, Cu, Pd, Pt, Rh foram investigadas através de cálculos de primeiros princípios. A estrutura destes compostos é considerada uma variante ternária da estrutura do tipo  $W_5Si_3$ . Tres possibilidades de ordenamento da estrutura foram investigadas neste trabalho. As duas estruturas nas quais os átomos de Sb e T' se alternam ao longo de cadeias paralelas ao eixo c da estrutura são as mais estáveis. Os parâmetros estruturais calculados estão em acordo com os valores experimentais. Tanto efeitos eletrônicos como de tamanho permitem explicar a estabilidade dos compostos ternários do tipo  $T_{10}Sb_5T'$ .

Palavras-chave: Cálculos DFT; Entalpias de formação; Variante ternária da estrutura W<sub>s</sub>Si<sub>3</sub>; Estrutura tipo Nb<sub>s</sub>Sn<sub>2</sub>Si.

### **I INTRODUCTION**

The existence of ternary compounds  $T_5X_2X'$ possessing the ternary D8<sub>m</sub> structure (Nb<sub>5</sub>Sn<sub>2</sub>Si-type, tl32, l4/mcm, N°140) [1] has been often reported in systems where T in an early transition metal (Ti, Zr, Hf, Nb, Ta), X a p element (Sn, Sb) and X' another p element (Al, Ga, Si, Ge). The Wyckoff positions of the atoms in the compound D8<sub>m</sub>-T<sub>5</sub>X<sub>2</sub>X' are the following: the T atoms occupy the 4b and 16k positions, the X atoms occupy the 8h while the X' atoms occupy the 4a positions. This ternary D8<sub>m</sub> structure was found in several systems of the type T–X–X' [1-17]. Tanaka et al. [10], Kleinke [13], Voznyak et al. [16] emphasized the arrangement of the p elements in the  $D8_m$  structure is controlled by size differences between X and X' elements.

In previous works [18-21], we performed *ab initio* calculations in order to determine the lattice parameters and the enthalpies of formation of D8<sub>m</sub>-Ti<sub>5</sub>Sn<sub>2</sub>Si [18], D8<sub>m</sub>-Ti<sub>5</sub>Sn<sub>2</sub>Ga [21], D8<sub>m</sub>-Nb<sub>5</sub>Sn<sub>2</sub>Ga [19] and of several D8<sub>m</sub>-Ti<sub>5</sub>Sb<sub>2</sub>X (X=AI, Ga, In, Si, Ge, Sn) [20] compounds. In these studies, we also concluded that the stability of these ternary D8<sub>m</sub> compounds is due to the size difference between X and X' elements ( $V_{sn} > V_{si}, V_{sn} > V_{Ga}$ ) or  $V_{sb} > V(AI, Ga, In, Si, Ge, Sn)$ . As a consequence, the distances X-X and T-T along the c axis are very short. Furthermore the stability of

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the compounds is favoured by an electronic factor due to a sufficient number of s and p electrons having the Fermi level near the bottom of the pseudo gap of the electronic density of states.

The aim of the present paper is to investigate the structural, thermodynamic, and electronic properties of another series of ternary compounds which possess a structure which is a variant-type of the  $W_sSi_3$  structure. These compounds have the formula  $T_{10}Sb_5T$  where T is an early transition metal (T= Ti, Zr, and Hf) and T' a transition metal of the 3d, 4d, or 5d rows, T' = V, Cr, Mn, Fe, Co, Ni, Cu, Rh, Pd, Pt. Several of them have been sinthetized and characterized [22-44]. Table I gathers the systems in which this structure has been found and characterized.

The more recent investigations concern the compounds Hf<sub>s</sub>Sb<sub>2</sub>Ru<sub>2</sub> and Zr<sub>s</sub>Sb<sub>2</sub>Ru<sub>2</sub> [39,44] which have be found to be superconductors. Table SI of the Supplement to this paper<sup>1</sup> gathers the lattice parameters of these compounds. Several ternary phase diagrams where the T<sub>10</sub>Sb<sub>5</sub>T' compound is stable have been investigated experimentally and isothermal sections at relatively low temperature have been drawn [22,28-30,32,34-36,39,41,43]. These systems are indicated by a star in Table 1. In most cases, a small deviation with respect to the stoichiometry  $T_{10}Sb_5T'$  is observed. The structure of these T<sub>10</sub>Sb<sub>5</sub>T' compounds is considered to be a substitution variant of the W<sub>5</sub>Si<sub>3</sub>-type structure. While in the stoichiometric  $D8_m$ - $T_{10}Sb_4T'_2$  compound, the 4a sites are occupied solely by the T' atoms, these sites are occupied by T' and Sb atoms in the  $T_{10}Sb_5T'$  compound. Note that in both structures, the 8h sites are occupied by Sb atoms while the 4b and 16k sites are occupied by the T atoms. At high temperature, a random distribution of Sb and T' atoms on the 4a sites is expected. It could be possible to calculate the enthalpy of formation of a partially random structure by using either supercells or special quasi-random structures. However, these calculations are long and sometimes difficult. Therefore, we have considered in the present work three possible ordered structures which have been proposed by Kleinke et al. [42]. The lattice parameters, enthalpies of formation, electronic densities of states of several  $T_{10}Sb_{z}T'$  compounds in these three structures have been computed. The results are also compared with those obtained for the  $D8_m$ - $T_{10}Sb_4T'_2$  compound.

#### 2 POSSIBLE ORDERED STRUCTURES OF T<sub>10</sub>Sb<sub>5</sub>T' COMPOUNDS

The unit cell of  $D8_m$ -T<sub>5</sub>Sb<sub>2</sub>T' contains 32 atoms that are situated in a layered arrangement along c axis. The layers z=1/4 and z=3/4 are occupied by T' (4a) and T (4b) atoms while the layers z=0 and z=1/2 are occupied by the T (16k) and Sb (8h) atoms. Figure 1a and 1e show the  $D8_m$  structure in the case of the Ti<sub>5</sub>Sb<sub>3</sub> and Ti<sub>5</sub>Sb<sub>2</sub>Co

**Table I.**  $T_{10}Sb_5T'$  ternary compounds with a structure which is a variant-type of the W<sub>5</sub>Si<sub>3</sub> structure

System	Composition	Ref
	Ti <sub>5</sub> Cu <sub>0.45</sub> Sb <sub>2.55</sub>	[22-25]
Ti-Cu-Sb*	Ti <sub>5</sub> CuSb <sub>2</sub>	[26]
	Ti <sub>s</sub> Cu <sub>I-x</sub> Sb <sub>2+x</sub>	[27]
	Ti <sub>5</sub> Ni <sub>0.57</sub> Sb <sub>2.43</sub>	[25]
<b>T:</b> NI: CL *	Ti <sub>5</sub> Ni <sub>0.45</sub> Sb <sub>2.55</sub>	[28]
Ti-Ni-Sb* — —	Ti <sub>5</sub> Ni <sub>0.45</sub> Sb <sub>2.55</sub>	[27]
	Ti <sub>5</sub> Ni <sub>0.45</sub> Sb <sub>2.55</sub>	[27]
	Ti <sub>5</sub> Co <sub>0.46</sub> Sb <sub>2.54</sub>	[25]
Ti-Co-Sb*	Ti <sub>5</sub> Co <sub>0.46</sub> Sb <sub>2.54</sub>	[29]
Ti-Co-Sb*	Ti <sub>5</sub> Co <sub>0.5</sub> Sb <sub>2.5</sub>	[27]
	Ti <sub>5</sub> Fe <sub>0.45</sub> Sb <sub>2.55</sub>	[30]
	Ti₅FeSb₂	[30]
11-Fe-3D*	Ti <sub>5</sub> Fe <sub>05</sub> Sb <sub>25</sub>	[27]
	Ti <sub>5</sub> Fe <sub>0.58</sub> Sb <sub>2.42</sub>	[25]
Ti-Mn-Sb	Ti <sub>5</sub> Mn <sub>0.45</sub> Sb <sub>2.55</sub>	[25]
	Ti <sub>5</sub> CrSb <sub>2</sub>	[31]
Ti-Cr-Sb	Ti <sub>5</sub> Cr <sub>05</sub> Sb <sub>25</sub>	[25]
Ti-Pd-Sb	Ti <sub>s</sub> Pd <sub>0</sub> sSb <sub>2</sub> s	[27]
	Ti <sub>s</sub> RhSb <sub>2</sub>	[27]
Ti-Rh-Sb	Ti <sub>s</sub> RhSb <sub>2</sub>	[27]
Ti-Ru-Sb	Ti <sub>s</sub> Ru <sub>o</sub> sb <sub>25</sub>	[27]
Zr-Cu-Sb*	Zr <sub>5</sub> Cu <sub>0.45</sub> Sb <sub>2.55</sub>	[32]
	Zr <sub>5</sub> Cu <sub>0.35</sub> Sb <sub>2.55</sub>	[33]
Zr-Ni-Sb*	Zr <sub>5</sub> Ni <sub>0.5</sub> Sb <sub>2.5</sub>	[34]
Zr-Co-Sb*	Zr <sub>5</sub> Co <sub>0.5</sub> Sb <sub>2.5</sub>	[35]
	Zr <sub>5</sub> Ni <sub>0.4</sub> Sb <sub>2.55</sub>	[33]
	Zr <sub>5</sub> Fe <sub>0.45</sub> Sb <sub>2.55</sub>	[33]
∠r-Fe-Sb*	Zr <sub>5</sub> Fe <sub>0.44</sub> Sb <sub>2.56</sub>	[36]
	Zr <sub>5</sub> Fe <sub>0.44</sub> Sb <sub>2.56</sub>	[37]
	Zr <sub>5</sub> Mn <sub>1-x</sub> Sb <sub>2+x</sub>	[38]
Zr-Mn-Sb*	Zr <sub>5</sub> Mn <sub>0.45</sub> Sb <sub>2.55</sub>	[32]
	Zr <sub>5</sub> Mn <sub>0.5</sub> Sb <sub>2.5</sub>	[39]
Zr-Cr-Sb	Zr <sub>5</sub> Cr <sub>0.49</sub> Sb <sub>2.51</sub>	[38]
Zr-Pd-Sb	Zr <sub>5</sub> Pd <sub>0.5</sub> Sb <sub>2.5</sub>	[27]
Zr-Rh-Sb	Zr <sub>5</sub> Rh <sub>0.55</sub> Sb <sub>2.5</sub>	[33]
Zr-Ru-Sb	Zr <sub>5</sub> Ru <sub>0.6</sub> Sb <sub>2.4</sub>	[40]
Zr-Pt-Sb*	Zr <sub>5</sub> PtSb <sub>2</sub>	[41]
	Hf <sub>10</sub> CuSb <sub>5</sub>	[42]
Hf-Cu-Sb*	Hf <sub>5</sub> Cu <sub>0.45</sub> Sb <sub>2.55</sub>	[43]
	Hf <sub>5</sub> Cu <sub>0.5</sub> Sb <sub>2.5</sub>	[27]
	Hf <sub>10</sub> NiSb <sub>5</sub>	[42]
Hf-Ni-Sb	Hf <sub>10</sub> Ni <sub>118</sub> Sb <sub>482</sub>	[38]
	Hf <sub>5</sub> Ni <sub>0</sub> ,Sb <sub>2</sub>	[27]
	Hf <sub>10</sub> CoSb <sub>5</sub>	[42]
Ht-Co-Sb	Hf,Co <sub>0</sub> ,Sb <sub>2</sub> ,	[27]

\*Systems for which isothermal sections have been drawn.

<sup>&</sup>lt;sup>1</sup>See DATA SUPPLEMENT to this paper in the same issue of this Journal.

System	Composition	Ref
Hf-Fe-Sb	Hf <sub>10</sub> FeSb <sub>5</sub>	[42]
	Hf <sub>10</sub> Fe <sub>1.54</sub> Sb <sub>4.46</sub>	[42]
	Hf <sub>4.929</sub> Fe <sub>0.67</sub> Sb <sub>2.33</sub>	[27]
Hf-Mn-Sb —	Hf <sub>10</sub> MnSb <sub>5</sub>	[42]
	Hf <sub>10</sub> Mn <sub>1.16</sub> Sb <sub>4.84</sub>	[42]
Hf-Cr-Sb	Hf <sub>10</sub> CrSb <sub>5</sub>	[42]
Hf-V-Sb	Hf <sub>10</sub> VSb <sub>5</sub>	[42]
	Hf <sub>10</sub> V <sub>0.8</sub> Sb <sub>5.2</sub>	[42]
	Hf <sub>10</sub> V <sub>0.88</sub> Sb <sub>5.12</sub>	[42]
Hf-Pd-Sb	$Hf_5Pd_{0.5}Sb_{2.5}$	[27]
Hf-Rh-Sb	Hf <sub>s</sub> Rh <sub>0.5</sub> Sb <sub>.5</sub>	[27]
	Hf <sub>s</sub> Ru <sub>0.5</sub> Sb <sub>2.5</sub>	[27]
Hf-Ru-Sb	Hf <sub>5</sub> Ru <sub>0.6</sub> Sb <sub>2.4</sub>	[44]

Table 1. Continued...

\*Systems for which isothermal sections have been drawn.

compounds. The structure is build up of two different columns of atoms running parallel to the tetragonal c axis. In the case of D8m-  $Ti_5Sb_2Co$ , one column consists of Co (4a) centred square antiprisms (CoTi<sub>8</sub>), which share square faces. The other column consists of Ti (4b) centred Sb tetrahedra (TiSb<sub>4</sub>) which share edges. These two columns are interconnected by numerous Ti-Sb and Ti-Ti bonds. The Ti (4b) as well as the Co (4a) atoms form parallel linear chains with short interatomic distances c/2.

The possible ordered structures of  $T_{10}Sb_5T$  compound which have been suggested by Kleinke et al. [42] are presented in Figures Ib-d: These structures, designated in the following by their space groups, are:

(i) N°97 in which the T' and Sb atoms alternate along the chains. The T' and Sb atoms also alternate in the planes z=0 and z=1/2.



**Figure 1.** Structures of (a)  $D8_m$ - $Sb_3Ti_5$ , (b)  $N^\circ 97$ - $Co_2Sb_{10}Ti_{20}$ , (c)  $N^\circ 125$ - $Co_2Sb_{10}Ti_{20}$ , (d)  $N^\circ 124$ - $Co_2Sb_{10}Ti_{20}$ , (e)  $D8_m$ - $Co_4Sb_8Ti_{20}$ . Green and dark green circles: Ti atoms (4b and 16k Wyckoff positions of the  $D8_m$  structure), orange circles: Sb atoms (8h Wyckoff positions of the  $D8_m$ ), white circles: Co atoms (4a Wyckoff positions of the  $D8_m$ ).

- (ii) N°125 in which again the T' and Sb atoms alternate along the chains but the T' atoms are in the z=0 plane while the Sb atoms in the z=1/2 plane
- (iii) N°124 in which there are alternate chains of T' atoms and Sb atoms parallel to the c axis.

The characteristics of these three ordered structures are given in Table 2 in the case of the  $Ti_{10}Sb_5Co$  compound (the values of the lattice parameters and internal parameters have been calculated as indicated below) and in Table 3 in the case of  $Hf_{10}Sb_5V$  compound.

Some remarks can already be done. In the three structures  $Ti_{10}Sb_5Co$ , the Ti atoms (4b in the  $D8_m$  description) are in the middle of a tetrahedron formed by the Sb atoms (8h in the  $D8_m$  description). On the other hand, the T' and Sb atoms (4a in the  $D8_m$  des) form chains inside the antiprisms build with the Ti (16k in the  $D8_m$  description) atoms. These chains are similar in the cases of the N°97 and N°125 structures. Figures S1 and S2 of the Supplement present the chains of Co and Sb atoms, and the chains of Ti atoms.

The aim of the present work is to perform first principles calculations of the enthalpies of formation of the N°97, N°125, and N°124 structures for several  $Ti_{10}Sb_5T'$  compounds in order to obtain their relative stabilities.

#### **3 COMPUTATIONAL METHOD**

The density functional (DFT) calculations were performed with the Vienna ab initio simulation package (VASP) [45,46], making use of the projector augmented waves (PAW) technique [47,48]. The calculations include twelve valence electrons for Ti (3s<sup>2</sup> 3p<sup>6</sup> 3d<sup>3</sup> 4s<sup>1</sup>). A plane-wave cut-off energy of 350 eV for all the elements has been taken. For the generalized gradient approximation (GGA) exchange correlation functional, the Perdew-Berke-Erzenhof parameterization (PBE) [49] was used. For the Brillouin-zone integration, the Methfessel-Paxton [50] technique with a modest smearing of the one-electron levels (0.2 eV) was used. Care was taken so that a sufficient number of k points for the Brillouin-zone integration was chosen for each structure. A Gamma centred k-point grid was used for the hexagonal structures while a Monkhorst-Pack [51] grid was chosen for the other structures. The tetrahedron method is used in the case of the stable compounds and for the electronic density of states computations. In this later case, the number of k points has been slightly increased. Spin-polarized calculations have been performed in the case of compounds with Cr, Mn, Fe, and Co.

The energy of formation per atom,  $\ddot{A}_{F}E(T_{10/16}Sb_{5/16}T_{1/16})$ , of the  $T_{10/16}Sb_{5/16}T_{1/16}$  compound is obtained from Equation 1:

**Table 2.** Calculated lattice parameters and internal coordinates of the atoms of the  $D8_m$ , and ternary variants of the  $D8_m$  structures in the Ti-Sb-Co system

Compound	Structure	Lattice parameters	Atomic positions		
	D8 <sub>m</sub> ,		Sb (4a) 0, 0, 1/4		
Ti <sub>10</sub> Sb <sub>6</sub>	W₅Si3-type,	a= 10.5061 Å	Sb (8h) 0.16392, 0.66392, 0		
	tl32, l4/mmc,	c= 5.4717 Å	Ti (4b) 0, 1/2, 1/4		
	N°140		Ti (16k) 0.07705, 0.22764, 0		
			Co (2a) 0, 0, 0		
	tl32,		Sb (2b) 0, 0, 1/2		
Ti₁₀Sb₅Co	1422,	a = 10.4907 A	Ti (4c) 0, 1/2, 0		
	N°97	C= 3.2147 A	Sb (8j) 0.16383, 0.66383, -0.25		
			Ti (16k) 0.07489, 0.21761, -0.24		
			Co (2a) 1/4, 1/4, 0		
	tP32, P4/nbm N°125		Sb (2b) 1/4, 1/4, 1/2		
T: Sh Ca		a= 10.4798 Å	Ti (2c) 3/4, 1/4, 0		
11 <sub>10</sub> 30 <sub>5</sub> CO		c= 5.2243 Å	Ti (2d) 3/4, 1/4, ½		
			Sb (8m) 0.4139, 0.5861, 0.2532		
			Ti (16n) 0.6754, 0.5322, 0.2382		
			Co (2a) 0, 0, 1/4		
	+D22		Sb (2c) 1/2, 1/2; 1/4		
T: Sh Ca	LF 32,	a= 10.4553 Å	Ti (4f) 0, 1/2, 1/4		
11 <sub>10</sub> 30 <sub>5</sub> CO	P4/mcc,	c= 5.3089 Å	Sb (8m) 0.16141, 0.66672, 0		
	IN 124		Ti (8m) 0.06969, 0.20045, 0.000		
			Ti (8m) 0.42301, 0.72846, 0.0000		
	D8 <sub>m</sub> ,		Co (4a) 0, 0, 1/4		
T: Sh Ca	Nb <sub>5</sub> Sn <sub>2</sub> Si-type	a= 10.4767 Å	Sb (8h) 0.16413, 0.66413, 0		
11 <sub>10</sub> 30 <sub>4</sub> CO <sub>2</sub>	tl32,	c= 5.0569 Å	Ti (4b) 0, 1/2, 1/4		
	14/mmc, N°140		Ti (16k) 0.07052, 0.20120, 0		

Tecnol. Metal. Mater. Miner., São Paulo, v. 13, n. 1, p. 75-90, jan./mar. 2016

$$\Delta_{f} \mathbb{E} \left( \mathbb{T}_{10/16} \mathbb{Sb}_{5/16} \mathbb{T}_{1/16}' \right) = \mathbb{E}_{T}^{\min} \mathbb{10/16}^{Sb} \frac{5}{5/16} \mathbb{E}_{1/16}^{Sb} - \frac{5}{16} \mathbb{E}_{A^{7}-Sb}^{\min} - \frac{1}{16} \mathbb{E}_{T'}^{\min}$$
(1)

The E<sup>min</sup> are the minimum total energy of the compound (expressed per atom) and of the elements (hexagonal close-packed A3-T, rhomboedric A7-Sb and pure T' elements in their standard state at 298K. At T=0K and p=0 Pa, the formation enthalpy,  $\Delta_{\rm f} H (T_{10/16} {\rm Sb}_{5/16} T_{1/16})$ , equals

the calculated formation energy, when the zero-vibration contribution is ignored, since it is much smaller than the formation energy.

#### **4 RESULTS AND DISCUSSION**

First of all, the ground state in the limiting binaries Ti-Sb, Zr-Sb, and Hf-Sb for the composition  $x_T = 0.625$  has been calculated. The results are reported in Table 4.

**Table 3.** Calculated lattice parameters and internal coordinates of the atoms of the  $D8_m$ , and ternary variants of the  $D8_m$  structures in the Hf-Sb-V system

Compound	Structure	Lattice parameters	Atomic positions		
	D8 <sub>m</sub> ,		Sb (4a) 0, 0, 1/4		
	W₅Si₃-type,	a= 11.0440 Å	Sb (8h) 0.16621, 0.66621, 0		
пі <sub>10</sub> 30 <sub>6</sub>	tl32,	c= 5.7202 Å	Hf (4b) 0, ½, 1/4		
	14/mmc, N°140		Hf (16k) 0.07477, 0.22452, 0		
			V (2a) 0, 0, 0		
	tl32,	2- 11 0649 Å	Sb (2b) 1/2, 1/2, 0		
Hf <sub>i0</sub> Sb₅V	1422,	a = 11.0077 Å	Hf (4c) 0, 1/2, 0		
	N°97	C— 5.5637 A	Sb (8j) 0.66351, 0.16351, 0.25		
			Hf (16k) 0.07632, 0.21566, -0.23634		
			V (2a) ¼, ¼, 0		
	tP32, P4/nbm		Sb (2b) 1/4, 1/4, 1/2		
		a= 11.0495 Å	Hf (2c) 3/4, 1/4, 0		
	N°125	c= 5.6064 Å	Hf (2d) 3/4, 1/4, 1/2		
			Sb (8m) 0.33618, 0.83618, 0.25429		
			Hf (16n) 0.32543, 0.46597, 0.23645		
			V (2a) 0, 0, 1/4		
	<b>+</b> ₽30		Sb (2c) 1/2, 1/2; 1/4		
	LF 52,	a= 11.0297 Å	Hf (4f) 0, 1/2, 1/4		
	P4/mcc,	c= 5.6518 Å	Sb (8m) 0.16337, 0.664654, 0		
	IN <sup>-</sup> 124		Hf (8m) 0.07190, 0.20557, 0.000		
			HF (8m) 0.42244, 0.72416, 0.0000		
	D8 <sub>m</sub> ,		V (4a) 0, 0, 1/4		
	Nb <sub>5</sub> Sn <sub>2</sub> Si-type	a= 11.0506 Å	Sb (8h) 0.16283, 0.66283, 0		
ΠI <sub>10</sub> 30 <sub>4</sub> ν <sub>2</sub>	tl32,	c= 5.5640Å	Hf (4b) 0, ½, 1/4		
	14/mmc, N°140		Hf (16k) 0.07369, 0.20549, 0		

1000 $100$	Table 4. Enthalpies of f	ormation of T <sub>2</sub> Sb <sub>2</sub>	(T=Ti, Zr,	Hf) in the D8_	(W <sub>c</sub> Si <sub>s</sub> -type)	, D8 <sub>°</sub> (Mn <sub>s</sub> Si,	-type), Yb <sub>c</sub> Sb	o,-type and Y,B	i,-type structu
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System	Composition	Structure	$\Delta_{\!{ m f}}{ m H}$ (kJ/mol of atoms) Non magnetic
		D8 <sub>m</sub>	-44.15
T: 5h	T: 5h	D8 <sub>8</sub>	-45.82
11-30	11,503	oP32, Yb <sub>5</sub> Sb <sub>3</sub> -type	-48.37
		oP32, Y <sub>5</sub> Bi <sub>3</sub> -type	-47.82
		D8 <sub>m</sub>	-62.95
Zr-Sb	7r Sh	D8 <sub>8</sub>	-64.92
	ZI 503	oP32, Yb <sub>5</sub> Sb <sub>3</sub> -type	-65.70
		oP32, Y <sub>5</sub> Bi <sub>3</sub> -type	-65.90
		D8 <sub>m</sub>	-41.19
	LIFCP	D8 <sub>8</sub>	-41.03
HI-30	HI <sub>5</sub> 30 <sub>3</sub>	oP32, Yb <sub>5</sub> Sb <sub>3</sub> -type	-44.88
		oP32, Y <sub>5</sub> Bi <sub>3</sub> -type	-44.93

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The results are in agreement with the experimental information: the Yb<sub>5</sub>Sb<sub>3</sub> structure type is the most stable in the case of the Ti<sub>5</sub>Sb<sub>3</sub> compound while the Y<sub>5</sub>Bi<sub>3</sub> structure type is the more stable in the cases of Hf<sub>5</sub>Sb<sub>3</sub> compounds. In the case of the Zr<sub>5</sub>Sb<sub>3</sub> compound, Garcia and Corbett [52] found that off stoichiometric D8<sub>8</sub>-Zr<sub>5</sub>Sb<sub>3+x</sub> is stable and that the oP32-Y<sub>5</sub>Bi<sub>3</sub>-type is stable at high temperature. Our calculations indicate that the Zr<sub>5</sub>Sb<sub>3</sub> in the Y<sub>5</sub>Bi<sub>3</sub>-type structure is 1kJ/mol of atoms more stable than the D8<sub>8</sub> structure. The calculated lattice parameters are in good agreement with the experimental data (Table 5).

For each investigated ternary system, the enthalpies of formation of the ordered compounds in structures  $D8_m$ , N°97, N°125, and N°124 are reported in Table 6. These results show that the N°97 and N°125 structures have very close enthalpies of formation, and that the enthalpy of formation of the N°124 structure is clearly less negative.

We have presented these results along sections in two cases:  $Ti_5Sb_3$ - $Ti_5Co_3$  (Figure 2) and  $Hf_5Sb_3$ - $Hf_5V_3$  (Figure 3).

The ground state in each limiting binary T-T' for  $x_r = 0.625$  molar fraction has been calculated and is indicated on the figures. These figures make possible getting some indication about the relative stability of the various compounds along the section. The ternary D8\_ structure T<sub>s</sub>Sb<sub>b</sub>T' appears stabilized with respect to the mechanical mixture of  $D8_m$ -T<sub>5</sub>Sb<sub>3</sub> and  $D8_m$ -T<sub>5</sub>T'<sub>3</sub>. This is due to the fact that the T' atoms possess a size which is lower than the one of the T atoms (T=Ti, Zr, Hf). Considering now, the section  $D8_{m}$ -T<sub>s</sub>Sb<sub>2</sub> to  $D8_{m}$ -T<sub>s</sub>Sb<sub>2</sub>T', one observes again that the structures  $N^{\circ}97$  and  $N^{\circ}125$  have very similar enthalpies of formation. There is an important stabilization of the  $N^{\circ}97-Ti_{10}Sb_{5}T'$  and  $N^{\circ}125-Ti_{10}Sb_{5}T'$  structure with respect to the mechanical mixture of D8\_-T\_Sb, and D8\_-T\_Sb,T' (see Figure 2 and 3). This stabilization decreases from Cu to Mn in the 3d series (see Table 6). The enthalpy of formation of the N°I24 structure is less negative or of the same order of magnitude than the mean value of the formation enthalpies of D8<sub>m</sub>-T<sub>5</sub>Sb<sub>3</sub> and D8<sub>m</sub>-T<sub>5</sub>Sb<sub>5</sub>T'.

For some of the investigated system, the experimental and calculated values of the lattice parameters are displayed on Figures 4 and 5. The lattice parameters calculated in the N°97 and N°125 ordered structures are in good agreement with the experimental values in the case of  $Ti_{10}Sb_5Co$ . In the case of  $Hf_{10}Sb_5V$ , the calculated values of the c parameter are in good agreement with the experimental ones, however the calculated values of the a parameter are clearly greater



Figure 2. Calculated values of the enthalpies of formation of the compounds along the section  $Ti_sSb_{1}-Ti_sCo_{2}$ .



Figure 3. Calculated values of the enthalpies of formation of the compounds along the section  $Hf_{s}Sb_{3}-Hf_{s}V_{3}$ .

i		1	5 3' 5 3'	5 3		
Compound	Structure		a [nm]	b [nm]	c [nm]	Reference
Ti <sub>s</sub> Sb <sub>3</sub>	Yb <sub>5</sub> Sb <sub>3</sub> , oP32	Exp.	1.02182	0.83432	0.71748	[30]
	Pnma, N°62	Exp.	1.02173	0.83281	0.71459	[53]
		Calc.	1.01777	0.83721	0.71983	This work
$Zr_{5}Sb_{3}$	Y₅Bi₃, oP32	Exp.	0.7465	0.8801	1.0865	[52]
	Pnma, N°62	Exp.	0.7465	0.8801	1.0865	[54]
		Calc.	0.7528	0.889	1.0966	This work
Zr <sub>5</sub> Sb <sub>3</sub>	Mn₅Si₃, hP16	Exp.	0.8422	0.8422	0.5696	[55,56]
	P63/mcm, N°193	Exp.	0.8488	0.8488	0.58	[57]
		Calc.	0.8515	0.8515	0.5797	This work
Hf <sub>5</sub> Sb <sub>3</sub>	Y₅Bi₃, oP32	Exp.	0.74075	0.8718	1.0736	[58]
	Pnma, N°62	Calc.	0.7443	0.8841	1.0786	This work

Table 5. Experimental and calculated lattice parameters of Ti<sub>2</sub>Sb<sub>2</sub>, Zr<sub>2</sub>Sb<sub>2</sub>, and Hf<sub>2</sub>Sb<sub>2</sub>

<b>Table 6.</b> Calculated values of the formation entitalpies of $D0^{-1}$ is $30^{-1}$ , and $1^{-1}$ , $10^{-1}$ ,
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			$\Delta_{f}H$	$\Delta_{\mathbf{f}}\mathbf{H}$
System	Composition	Structure	(kJ/mol of atoms)	(kJ/mol of atoms)
			Non magnetic	Spin pol.
Ti-Cu-Sb	Ti <sub>10</sub> CuSb <sub>5</sub>	N° <b>97</b>	-47.03	
		N°125	-47.00	
		N°124	-40.43	
	Ti <sub>10</sub> Cu <sub>2</sub> Sb <sub>4</sub>	D8 <sub>m</sub>	-38.16	
Ti-Ni-Sb	Ti <sub>10</sub> NiSb <sub>5</sub>	N°97	-49.71	
		N°125	-49.72	
		N°124	-42.98	
	Ti <sub>10</sub> Ni <sub>2</sub> Sb <sub>4</sub>	D8 <sub>m</sub>	-44.60	
Ti-Co-Sb	Ti <sub>10</sub> CoSb <sub>5</sub>	N°97	-48.09	-48.10
		N°125	-47.87	-47.91
		N°124	-42.59	-42.60
	Ti <sub>10</sub> Co <sub>2</sub> Sb <sub>4</sub>	D8 <sub>m</sub>	_44.41	-44.46
Ti-Fe-Sb	Ti <sub>10</sub> FeSb <sub>5</sub>	N°97	-46.30	-46.57
		N°125	-45.98	-46.38
		N°124	-41.77	-41.78
	Ti <sub>10</sub> Fe <sub>2</sub> Sb <sub>4</sub>	D8 <sub>m</sub>	-42.20	-42.09
Ti-Mn-Sb	Ti <sub>10</sub> MnSb <sub>5</sub>	N°97	-45.32	-45.22
		N°125	-44.90	-44.97
		N°124	-40.76	-40.89
	Ti <sub>10</sub> Mn <sub>2</sub> Sb <sub>4</sub>	D8 <sub>m</sub>	-40.45	-40.58
Ti-Cr-Sb	Ti <sub>10</sub> CrSb <sub>5</sub>	N°97	-42.93	
		N°125	-42.64	
		N°124	-38.46	
	Ti <sub>10</sub> Cr <sub>2</sub> Sb <sub>4</sub>	D8 <sub>m</sub>	-35.71	
Ti-Pt-Sb	Ti <sub>10</sub> PtSb <sub>5</sub>	N°97	-55.24	
		N°125	-55.19	
		N°124	-50.84	
	Ti <sub>10</sub> Pt <sub>2</sub> Sb <sub>4</sub>	D8 <sub>m</sub>	-58.34	
Ti-Rh-Sb	Ti <sub>10</sub> RhSb <sub>5</sub>	N°97	-52.08	
		N°125	-52.22	
		N°124	-46.19	
	Ti <sub>10</sub> Rh <sub>2</sub> Sb <sub>4</sub>	D8 <sub>m</sub>	-50.5 l	
Zr-Cu-Sb	Zr <sub>10</sub> CuSb <sub>5</sub>	N°97	-62.05	
		N°125	-62.00	
		N°124	-57.01	
	$Zr_{10}Cu_2Sb_4$	D8 <sub>m</sub>	-50.9 l	
Zr-Ni-Sb	Zr <sub>10</sub> NiSb <sub>5</sub>	N°97	-63.48	
		N°125	-63.65	
		N°124	-58.83	
	$Zr_{10}Ni_2Sb_4$	D8 <sub>m</sub>	-55.32	
Zr-Co-Sb	Zr <sub>10</sub> CoSb <sub>5</sub>	N° <b>97</b>	-61.46	
	· –	N°125	-61.29	
		N°124	-57.70	
	Zr <sub>10</sub> Co <sub>2</sub> Sb <sub>4</sub>	D8 <sub>m</sub>	-53.52	

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#### Table 6. Continued...

			$\Delta_{f}\mathbf{H}$	$\Delta_{f}H$
System	Composition	Structure	(kJ/mol of atoms)	(kJ/mol of atoms)
			Non magnetic	Spin pol.
Zr-Fe-Sb	Zr <sub>10</sub> FeSb <sub>5</sub>	N°97	-59.59	-60.27
		N°125	-58.91	-59.99
		N°124	-56.21	-56.21
	$Zr_{10}Fe_2Sb_4$	D8 <sub>m</sub>	-50.24	-49.47
Zr-Pt-Sb	Zr <sub>10</sub> PtSb <sub>5</sub>	N°97	-70.84	
		N°125	-70.78	
		N°124	-67.60	
	$Zr_{10}Pt_{2}Sb_{4}$	D8 <sub>m</sub>	-72.47	
Hf-Cu-Sb	Hf <sub>I0</sub> CuSb₅	N°97	-44.83	
		N°125	-44.80	
		N°124	-39.67	
	Hf <sub>10</sub> Cu <sub>2</sub> Sb <sub>4</sub>	D8 <sub>m</sub>	-37.53	
Hf-Ni-Sb	Hf₁₀NiSb₅	N°97	-47.17	
		N°125	-47.23	
		N°124	-42.39	
	Hf <sub>10</sub> Ni <sub>2</sub> Sb <sub>4</sub>	D8 <sub>m</sub>	-43.57	
Hf-V-Sb	Hf <sub>10</sub> VSb <sub>5</sub>	N°97	-41.40	
		N°125	-41.24	
		N°124	-37.91	
	$Hf_{10}V_2Sb_4$	D8 <sub>m</sub>	-36.22	



**Figure 4.** Calculated and experimental [25,27,29] values of the lattice parameters of the D8<sub>m</sub> compounds along the section  $Ti_sSb_3$ - $Ti_sCo_3$ .

than the experimental ones. One explanation could be that the lattice parameters calculated in the GGA approximation are the more often greater than the experimental values. The values of the lattice parameters calculated for all compounds investigated in the present work are reported in the *Supplement* to this paper.

The distances between the atoms may be calculated and are reported in Table 7 in the case of the  $D8_m$ -Ti<sub>10</sub>Sb<sub>4</sub>Co<sub>2</sub> compound and in Tables 8-10 in the cases of the three possible ordered structures of Ti<sub>10</sub>Sb<sub>5</sub>Co.



**Figure 5.** Calculated and experimental [41] values of the lattice parameters of the  $D8_m$  compounds along the section  $Hf_5Sb_3$ - $Hf_5V_3$ .

One may observe that the interatomic distances are the same in the N°97 and N°125 structures. Indeed as indicated in section 2, the only difference between the two structures is the positions of the Co and Sb atoms in the planes z=0 and z=1/2 (see Figures S1 and S2 of the *Supplement*). In another hand, the lattice parameters and the interatomic distances are different in the N°124 structure. In fact, the N°124 structure may be considered as the weighted sum of the D8<sub>m</sub>-Ti<sub>5</sub>Sb<sub>3</sub> and D8<sub>m</sub>-Ti<sub>5</sub>Sb<sub>2</sub>Co structures. This is true for the lattice parameters and for the atomic distances. As a

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		,	5-2	-F					
	Co (4a)			Sb (8h)	Ti (4b)		Ti (16k)		
Co (4a)	2	2.530 Å					8	2.568 Å	
Sb (8h)					2	2.742 Å	8	2.808-3.058 Å	
Ti (4b)			4	2.742 Å	2	2.530 Å	8	3.458 Å	
Ti (16k)	2	2.568 Å	4	2.808-3.058 Å	2	3.458 Å	5	2.930-3.186 Å	

Table 7. Bond lengths in the ternary D8 -Ti<sub>c</sub>Sb<sub>2</sub>Co compound

Table 8. Bond lengths in the ternary N°97-Ti10Sb5Co compound

		Co (2a)		Sb (2b)		Sb (8j)		Ti (4c)		Ti (16k)
Co (2a)			2	2.607 Å					8	2.719 Å
Sb (2b)	2	2.607 Å							8	2.769 Å
Sb (8j)							2	2.758 Å	8	2.797-3.081 Å
Ti (4c)					4	2.758 Å	2	2.607 Å	8	3.311-3.351 Å
Ti (16k)	Ι	2.719 Å	Ι	2.769 Å	4	2.797-3.081 Å	2	3.311-3.351 Å	7	2.955-3.414 Å

Table 9. Bond lengths in the ternary N°125-Ti10Sb5Co compound

		Co (2a)		Sb (2b)		Sb (8k)	Т	'i (2c), Ti (2d)		Ti (16l)
Co (2a)			2	2.612 Å					8	2.715 Å
Sb (2b)	2	2.612 Å							8	2.773 Å
Sb (8k)							Ι	2.766 Å	8	2.799-3.078 Å
							I	2.750 Å		
Ti (2c)					4	2.766 Å	2	2.612 Å	8	3.302 Å
Ti (2d)					4	2.750 Å	2	2.612 Å	8	3.351 Å
Ti (16l)	Ι	2.715 Å	Ι	2.773 Å	4	2.799-3.078 Å	I	3.302 Å	7	2.939-3.462 Å
							I	3.351 Å		

Table 10. Bond lengths in the ternary N°124-Ti10Sb5Co compound

		Co (2a)		Sb (2c)		Sb (8m)		Ti (4f)		Ti (16l)
Co (2a)	2	2.654 Å	2	2.654 Å					8	2.585 Å
Sb (2c)	2	2.654 Å							8	2.849 Å
Sb (8k)							2	2.765 Å	4	2.786-3.146 Å
Ti (4f)					4	2.765 Å	2	2.654 Å	8	3.236-3.479 Å
Ti (16l)	2	2.585 Å	2	2.849 Å	4	2.786-3.146 Å	2	3.236, 3.479 Å	6	3.028-3.473 Å

consequence, the enthalpy of formation of N°124-Ti<sub>10</sub>Sb<sub>5</sub>Co is in the middle of the line joining the enthalpies of formation of D8<sub>m</sub>-Ti<sub>5</sub>Sb<sub>3</sub> and D8<sub>m</sub>-Ti<sub>5</sub>Sb<sub>2</sub>Co structures. This observation can be made in all other systems we have studied in the present work.

The Bader charges for the series  $D8_m$ -Ti<sub>5</sub>Sb<sub>2</sub>T' and N°97- Ti<sub>5</sub>Sb<sub>2.5</sub>T'<sub>0.5</sub> (T' = Co, Ni) have been computed. The transfer from the Ti atoms to the late transition metal is in these cases of about 1.5 e<sup>-</sup>, leading in these cases to a filled d band of Co and Ni.

At high temperature, one expects a random distribution of the Sb and T' atoms along the chains corresponding to the 4a sites of the  $D8_m$  structure. As a consequence the formation enthalpy of the disordered phase will be less negative than the ones of N°97 and N°125 but the stability will be maintained by the mixing entropy of Sb and T'.

#### **5 CONCLUSION**

In the present work, we have investigated ternary compounds  $T_{10}Sb_5T'$  where T=Ti, Zr, Hf, and T' a 3d, or 4d, or 5d transition metal of the end of the series. We have calculated the enthalpies of formation of  $D8_m$ - $T_5Sb_2T'$  and of ordered structures of composition  $T_{10}Sb_5T'$ . We have shown that the structures where the T' and Sb atoms alternate along the chains formed by the 4a sites of the  $W_5Si_3$  structure are the most stable structures (structures N°97 and N°125). These structures are very similar, the only difference is the positions of T' and Sb atoms in the plans z=0 and z=1/2. On the contrary, the structure where chains formed by T' atoms and Sb atoms solely is the less stable one (structure N°124).

The stability of the  $T_{10}Sb_5T'$  compound is due to several factors. First, even if the  $D8_m$ -structure of  $Ti_5Sb_3$ ,

 $Zr_sSb_3$ , and  $Hf_sSb_3$  is not the ground state in the binaries Ti-Sb, Zr-Sb, and Hf-Sb, the energy difference between the D8<sub>m</sub> and the ground state structure is not very important (less than 4.2 kJ/mol of atoms). Secondly, the D8<sub>m</sub>-T<sub>s</sub>Sb<sub>2</sub>T' compound in the Nb<sub>s</sub>Sn<sub>2</sub>Si-type is stabilized with respect the two D8<sub>m</sub>-T<sub>s</sub>Sb<sub>3</sub> and D8<sub>m</sub>-T'<sub>s</sub>Sb<sub>3</sub> compounds, due to the size of the T' atom which is smaller than the one of the T atoms. In previous publications, this fact was shown in the case of T<sub>s</sub>Sb<sub>2</sub>X compounds where X is a p element. Finally the composition T<sub>10</sub>Sb<sub>5</sub>T' is favoured by an electronic factor, which is a complete or partial filling of the d band of the T' transition metal. At high temperature, the Sb and T' atoms present a random distribution on the 4a chains of the  $W_5Si_3$ -type structure. The formation enthalpy is therefore less negative than the one of the N°97 or N°125. The corresponding increase of the Gibbs energy is compensated by a configurational entropy due to the random mixing of Sb and T' atoms along the chains.

This study is a first step in the investigation of isothermal sections in the T-Sb-T' systems. Such studies require the knowledge of the formation enthalpies of all compounds present in the system.

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Received: 18 Jan. 2016 Accepted: 21 Jan. 2016 **Supplement.** Structural stability of ternary antimonides  $T_{10}Sb_5T'$  (T=Ti, Zr, Hf, T'=V, Cr, Mn, Fe, Co, Ni, Cu, Pd, Pt, Rh) **Suplemento.** Estabilidade estrutural de compostos ternários de antimonio  $T_{10}Sb_5T'$  (T=Ti, Zr, Hf, T'=V, Cr, Mn, Fe, Co, Ni, Cu, Pd, Pt, Rh)

			$\Delta_{f}H$	$\Delta_{f}H$
System	Composition	Structure	(kJ/mol of atoms)	(kJ/mol of atoms)
			Non magnetic	Spin pol
	Ti₁₀CuSb₅	N°97	-47.027	
		N°125	-47.002	
		N°124	-40.432	
Ti-Cu-Sb	Ti <sub>10</sub> Cu <sub>2</sub> Sb <sub>4</sub>	D8 <sub>m</sub>	-38.158	
	Ti <sub>10</sub> Cu <sub>6</sub>	D8 <sub>m</sub>	-0.708	
	GS CI Ib-Ti <sub>2</sub> Cu	CIIb-Ti <sub>2</sub> Cu	-11.686	
	+BII-TiCu	B11-TiCu	-13.389	
	Ti <sub>10</sub> NiSb₅	N° <b>97</b>	-49.710	
		N°125	-49.721	
		N°124	-42.982	
Ti-Ni-Sb	Ti <sub>10</sub> Ni <sub>2</sub> Sb <sub>4</sub>	D8 <sub>m</sub>	-44.596	
	Ti <sub>10</sub> Ni <sub>6</sub>	D8 <sub>m</sub>	-26.323	
	GS E9 <sub>3</sub> -Ti <sub>2</sub> Ni	E9 <sub>3</sub> -Ti <sub>2</sub> Ni	-27.049	
	+B27-TiNi	B27-TiNi	-38.776	
	Ti₁₀CoSb₅	N° <b>97</b>	-48.086	
		N°125	-47.869	
		N°124	-42.590	-48.095
Ti-Co-Sb	Ti <sub>10</sub> Co <sub>2</sub> Sb <sub>4</sub>	D8 <sub>m</sub>	-44.409	-47.911
	Ti <sub>10</sub> Co <sub>6</sub>	D8 <sub>m</sub>	-25.180	-42.602
	GS E9 <sub>3</sub> -Ti <sub>2</sub> Co	E9 <sub>3</sub> -Ti <sub>2</sub> Co	-27.683	-44.463
	+B2-TiCo	B2-TiCo	-36.571	
	Ti <sub>10</sub> FeSb <sub>5</sub>	N° <b>97</b>	-46.298	
		N°125	-45.984	-46.574
		N°124	-41.768	-46.380
Ti-Fe-Sb	Ti <sub>10</sub> Fe <sub>2</sub> Sb <sub>4</sub>	D8 <sub>m</sub>	-42.200	-41.782
	Ti <sub>10</sub> Fe <sub>6</sub>	D8 <sub>m</sub>	-13.240	-42.089
	GS E9 <sub>3</sub> -Ti <sub>2</sub> Fe	E9 <sub>3</sub> -Ti <sub>2</sub> Fe	-21.316	-13.659
	+B2-Ti-Fe	B2-Ti-Fe	-40.316	
	Ti <sub>10</sub> MnSb <sub>5</sub>	N° <b>97</b>	-45.3171	
		N°125	-44.9047	
		N°124	-40.7612	-45.2151
Ti-Mn-Sb	Ti <sub>10</sub> Mn <sub>2</sub> Sb <sub>4</sub>	D8 <sub>m</sub>	-40.45 I	-44.9660
	Ti <sub>10</sub> Mn <sub>6</sub>	D8 <sub>m</sub>	-I.9705	-40.8914
	GS A3-Ti	A3-Ti	0	-40.576
	+B2-TiMn	B2-TiMn	-23.702	
	Ti <sub>10</sub> PtSb <sub>5</sub>	N°97	-55.244	
		N°125	-55.190	
Ti Pr Sh		N°124	-50.840	
11-71-30	Ti <sub>10</sub> Pt <sub>2</sub> Sb <sub>4</sub>	D8 <sub>m</sub>	-58.339	
	Ti <sub>10</sub> Pt <sub>6</sub>	D8 <sub>m</sub>	-71.509	
	GS D8 <sub>8</sub> -Ti <sub>5</sub> Pt <sub>3</sub>	D8 <sub>8</sub> -Ti <sub>5</sub> Pt <sub>3</sub>	-75.673	

 Table SIa. Calculated values of the formation enthalpies of Ti based compounds

## Table SIa. Continued...

			$\Delta_{f}H$	$\Delta_{\mathbf{f}}\mathbf{H}$
System	Composition	Structure	(kJ/mol of atoms)	(kJ/mol of atoms)
			Non magnetic	Spin pol
	Ti₁₀RhSb₅	N°97	-52.002	
		N°125	-52.616	
		N°I24	-46.578	
Ti-Rh-Sb	Ti <sub>10</sub> Rh₂Sb₄	D8 <sub>m</sub>	-50.463	
	Ti <sub>10</sub> Rh <sub>6</sub>	D8 <sub>m</sub>	-52.500	
	GS CIIb-Ti₂Rh	C11b-Ti₂Rh	-63.131	
	+LI <sub>0</sub> -TiRh	LI <sub>0</sub> -TiRh	-75.847	

## Table SIb. Calculated values of the formation enthalpies of Zr based compounds

			$\Delta_{\mathbf{f}}\mathbf{H}$	$\Delta_{f}H$
System	Composition	Structure	(kJ/mol of atoms)	(kJ/mol of atoms)
			Non magnetic	Spin pol
	Zr <sub>10</sub> CuSb <sub>5</sub>	N°97	-62.047	
		N°125	-62.004	
		N°124	-57.012	
Zr-Cu-Sb	Zr <sub>10</sub> Cu <sub>2</sub> Sb <sub>4</sub>	D8 <sub>m</sub>	-50.910	
	Zr <sub>10</sub> Cu <sub>6</sub>	D8 <sub>m</sub>	-4.074	
	GS CI Ib-Zr <sub>2</sub> Cu	CIIb-Zr <sub>2</sub> Cu	-13.2936	
	$+Zr_7Cu_{10}$	oS68-Zr <sub>7</sub> Cu <sub>10</sub>	-15.7476	
	Zr <sub>10</sub> NiSb <sub>5</sub>	N°97	-63.480	
		N°125	-63.654	
		N°124	-58.834	
Zr-Ni-Sb	$Zr_{10}Ni_2Sb_4$	D8 <sub>m</sub>	-55.32I	
	Zr <sub>10</sub> Ni <sub>6</sub>	D8 <sub>m</sub>	-25.164	
	GS CI6-Zr <sub>2</sub> Ni	CI6-Zr <sub>2</sub> Ni	-31.299	
	+B33-ZrNi	B33-ZrNi	-44.028	
	Zr <sub>10</sub> CoSb <sub>5</sub>	N°97	-61.456	
		N°125	-61.294	
		N°124	-57.698	
Zr-Co-Sb	Zr <sub>10</sub> Co <sub>2</sub> Sb <sub>4</sub>	D8 <sub>m</sub>	-53.521	
	Zr <sub>10</sub> Co <sub>6</sub>	D8 <sub>m</sub>	-19.930	
	GS CI6-Zr <sub>2</sub> Co	CI6-Zr <sub>2</sub> Co	-24.540	
	+B27-ZrCo	B27-ZrCo	-31.389	
	Zr <sub>10</sub> FeSb <sub>5</sub>	N°97	-59.587	-60.272
		N°125	-58.914	-59.992
		N°124	-56.213	-56.213
Zr-Fe-Sb	$Zr_{10}Fe_2Sb_4$	D8 <sub>m</sub>	-50.235	-49.472
	Zr <sub>10</sub> Fe <sub>6</sub>	D8 <sub>m</sub>	-5.864	-5.981
	GS E9 <sub>3</sub> -Zr <sub>2</sub> Fe	E9 <sub>3</sub> -Zr <sub>2</sub> Fe	-11.649	
	+CI5-ZrFe2	C15-ZrFe2	-16.884	-26.980
	Zr <sub>10</sub> PtSb <sub>5</sub>	N°97	-70.840	
		N°125	-70.776	
Zr-Pt-Sb		N°124	-67.599	
	$Zr_{10}Pt_2Sb_4$	D8 <sub>m</sub>	-72.472	
	GS D8 <sub>m</sub> -Zr <sub>10</sub> Pt <sub>6</sub>	D8 <sub>m</sub> -Zr <sub>10</sub> Pt <sub>6</sub>	-79.127	

			$\Delta_{\mathbf{f}}\mathbf{H}$	$\Delta_{\mathbf{f}}\mathbf{H}$	
System	Composition	Structure	(kJ/mol of atoms)	(kJ/mol of atoms)	
			Non magnetic	Spin pol	
	Hf <sub>10</sub> CuSb₅	N°97	-44.832		
		N°125	-44.798		
		N°124	-39.670		
Hf-Cu-Sb	Hf <sub>10</sub> Cu₂Sb₄	D8 <sub>m</sub>	-37.531		
	Hf <sub>10</sub> Cu <sub>6</sub>	D8	-1.827		
	GS CI Ib-Hf,Cu	CI Ib-Hf,Cu	-15.639		
	$+Hf_{2}Cu_{10}$	oS68-Hf <sub>7</sub> Cu <sub>10</sub>	-16.903		
	Hf <sub>10</sub> NiSb₅	N°97	-47.169		
		N°125	-47.228		
		N°124	-42.387		
Hf-Ni-Sb	Hf <sub>10</sub> Ni <sub>2</sub> Sb <sub>4</sub>	D8 <sub>m</sub>	-43.571		
	Hf <sub>10</sub> Ni <sub>6</sub>	D8 <sub>m</sub>	-27.959		
	GS C16-Hf,Ni	CI6-Hf,Ni	-32.916		
	+B33-HfNi	B33-HfNi	-50.007		
	Hf <sub>10</sub> VSb <sub>5</sub>	N° <b>97</b>	-41.402		
		N°125	-41.240		
		N°124	-37.910		
Hf-V-Sb	$Hf_{10}V_2Sb_4$	D8 <sub>m</sub>	-36.217		
	Hf <sub>10</sub> V <sub>6</sub>	D8	19.502		
	GS		0		
	C14-	CI4-HfV2	0.8829		

Table SIc.	Calculated	values of th	e formation	enthalpies o	of Hf based	compounds



**Figure S1.** Chains of Sb and Co atoms in the cases of N°97 and N°125 structures of  $Ti_{10}Sb_{s}Co$ . The Ti atoms form antiprisms which share faces.



Figure S2. Ti 4 centred Sb tetrahedra which share edges. This figure is the same in the N°97, N°125 and N°124 structures.